

# A fast charge-dependent atom-pairwise dispersion correction for DFTB3

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## Abstract

Organic electronic materials remarkably illustrate the importance of the "weak" dispersion interactions that are neglected in the most cost-efficient electronic structure approaches. This work introduces a fast atom-pairwise dispersion correction, dDMC that is compatible with the most recent variant of Self-consistent charge Density functional tight binding (SCC-DFTB). The emphasis is placed on improving the description of  $\pi$ - $\pi$  stacked motifs featuring sulfur-containing molecules that are known to be especially challenging for DFTB. Our scheme relies upon the use of Mulliken charges using minimal basis set that are readily available from the DFTB computations at no additional cost. The performance and the efficiency of the dDMC correction are validated on series of examples targeting energies, geometries and molecular dynamic trajectories.

## Introduction

The development of electronic devices based on  $\pi$ -conjugated polymers and oligomers<sup>1-3</sup> is driven by the opportunity to achieve novel functionalities (e.g. mechanical flexibility, transparency, impact resistance)<sup>4-7</sup> at reduced fabrication costs. The performance of such organic devices depends

22 heavily upon the organization of  $\pi$ -conjugated molecules or chains at the molecular level<sup>3,8</sup> and  
23 upon the electronic structure mirrored by the wavefunction. In this context, insightful structure-  
24 property relationships can be exploited if quantum chemistry is used concurrently with experiment.  
25 The main attractive interactions occurring between  $\pi$ -conjugated moieties arise from van der Waals  
26 forces that decay as  $R^{-6}$  at large intermolecular distances. The central role of computational  
27 approaches is hence to achieve an accurate description of London dispersion and establish how to  
28 fine-tune the relative displacements or orientations between  $\pi$ -conjugated cores. Despite their  
29 omnipresence, van der Waals interactions are not accounted for by standard semi-local and hybrid  
30 density functionals<sup>9-11</sup> that provide a practical balance of accuracy and computational cost  
31 unmatched by other methods. Over the last decade, tremendous efforts have been devoted to  
32 improving the description of dispersion forces within the DFT framework.<sup>12,13</sup> The most extensively  
33 used approaches consist in adding *a posteriori* an atom pairwise energy correction term (*vide*  
34 *infra*).<sup>14-18</sup> The various available pairwise schemes differ in the way the dispersion coefficients are  
35 obtained. For instance, Grimme's popular DFT-D is based only on pre-tabulated values,<sup>17,19,20</sup> the  
36 XDM model from Becke and Johnson computes the dispersion coefficients from the exchange-hole  
37 dipole moment,<sup>21,22</sup> Tkatchenko and Scheffler's vdW-TS connects the dispersion coefficients to the  
38 size of the atom in the molecule,<sup>15</sup> while their latest variant also accounts for the many-body  
39 physics.<sup>23,24</sup> Closer to the present context, Steinmann *et al.* formulated a classical<sup>25</sup> and density-  
40 dependent dispersion correction (dDsC),<sup>18,26-28</sup> which simplifies the computation of the XDM and  
41 exploits Hirshfeld (overlap) populations<sup>29</sup> to distinguish non-bonded regions from bonded atom  
42 pairs, eliminating the correction at covalent distances.<sup>18</sup>

43 The DFT framework as used in practice is convenient and efficient albeit restricted to systems made  
44 of few hundred atoms only. This limitation prevents the modeling of large-scale organic molecular  
45 materials. In comparison, tight binding and other semi-empirical approaches are capable of  
46 producing molecular geometries and energetics at dramatically reduced computational costs.<sup>30</sup> In  
47 particular, the Self-Consistent Charge Density Functional Tight-Binding (SCC-DFTB)<sup>31</sup> scheme,  
48 rooted within the DFTB method developed by Seifert *et al.*<sup>32,33</sup> as well as its most recent DFTB3

49 variant,<sup>34-36</sup> provide valuable insights and have already unraveled complex problems.<sup>37-40</sup> In the  
50 context of organic electronic materials, the novel parameterization for (bio)organic molecules (so  
51 called 3OB) is especially relevant as it restores the proper qualitative behavior<sup>41,42</sup> for molecules  
52 involving non-covalently bound sulfur atoms that were poorly described by the previous MIO11  
53 parameter set.<sup>43-45</sup> Yet, SCC-DFTB suffers from the same deficiency as DFT functionals and does  
54 not account for dispersion interactions.<sup>9,46</sup> In this work, we propose a dispersion correction tailored  
55 for SCC-DFTB but inspired from the density-dependent dDsC correction. The proposed model is  
56 called dDMC due to its dependence on Mulliken charges<sup>47</sup> that are readily available from a SCC-  
57 DFTB computation. As such, dDMC does not require any additional information and is  
58 computationally very cheap. Alternative dispersion energy corrections adapted to SCC-  
59 DFTB/DFTB3 approaches exist. Elstner *et al.* proposed a method suitable for biological system that  
60 is based on the Slater-Kirkwood effective number of electrons.<sup>9</sup> Zhechkov *et al.* used the Universal  
61 Force Field London coefficients to correct the SCC-DFTB energy.<sup>46</sup> Rezac *et al.*, introduced the  
62 more sophisticated D3H4<sup>48</sup> method, which corrects for dispersion interaction using Grimme's D3<sup>17</sup>  
63 correction and improve the description of interactions involving hydrogen atoms. More recently,  
64 Grimme proposed a new parameterization of D3 specific for the DFTB3/3OB.<sup>49</sup> However, all these  
65 schemes have been parameterized and validated on biological systems with no specific  
66 consideration of typical  $\pi$ - $\pi$  stacked molecules characteristics of organic electronics. The challenges  
67 associated with the modeling of these systems involve overcoming the interplay arising from the  
68 poor description of both the sulfur-containing moieties<sup>41,45</sup> inherent to the DFTB parameters and the  
69 vdW interactions. The pragmatic dispersion correction proposed herein aims at providing efficiently  
70 reliable energies, geometries and molecular dynamic trajectories for sulfur-containing organic  
71 complexes. The next section describes the theoretical aspects of the dDMC that is followed by its  
72 validation.

73 **Theory**

74 dDMC is an *a posteriori* pairwise dispersion correction that adjusts the idea behind dDsC<sup>18,25,26</sup> to  
 75 the simpler and less computationally expensive DFTB scheme.

76 The general approach to compute the dispersion energy is

$$77 \quad E_{disp} = \sum_i^{N-1} \sum_{j>i}^N f_d(R_{ij}) \frac{C_{6,ij}}{R_{ij}^6} \quad (1)$$

78 The indexes  $i$  and  $j$  run over all the nuclei,  $R_{ij}$  is the internuclear distance,  $C_{6,ij}$  is the dispersion  
 79 coefficient associated with the interaction between the atom  $i$  and the atom  $j$ ,  $f_d$  is a function that  
 80 damps the correction at short internuclear distances that are better described by the DFTB  
 81 Hamiltonian. The commonly used density-dependent schemes (e.g., XDM, dDsC) compute atomic  
 82 dispersion coefficients from partitioning functions such as the Hirshfeld scheme. The same  
 83 Hirshfeld partitioning is also used in the sophisticated damping function of dDsC. The  
 84 simplification in dDMC aims at avoiding the computation of: i) integrals inherent to the Hirshfeld  
 85 partitioning and ii) local density derivatives (i.e., the XDM) that are more demanding than SCC-  
 86 DFTB itself. The Hirshfeld partitioning defines a weighting function:

$$87 \quad w_i = \frac{\rho_i^{free}(\vec{r})}{\sum_j \rho_j^{free}(\vec{r})} \quad (3)$$

88 where  $\rho_i^{free}(\vec{r})$  represents the electron density associated with the free atom  $i$  while  $\rho(\vec{r})$  is the  
 89 molecular electron density. The  $j$  index runs over all the atoms in the molecule.

90 One of the central quantities, on which density dependent dispersion corrections are based, is the  
 91 estimate of the size of the atom in a molecule. In particular, the ratio between the volume of the  
 92 AIM and the free atom:

$$93 \quad \frac{V_i^{aim}}{V_i^{free}} = \left( \frac{\int r^3 w_i(\vec{r}) \rho(\vec{r}) d^3\vec{r}}{\int r^3 \rho_i^{free}(\vec{r}) d^3\vec{r}} \right) \quad (4)$$

94 can be conveniently approximated by Equation (5).

95 
$$\frac{V_i^{aim}}{V_i^{free}} \approx \frac{N_i}{Z_i} \quad (5)$$

96 where  $N_i$  and  $Z_i$  are the Mulliken electronic population for the atom in the molecule and the  
 97 number of electrons for the free atom  $i$ . This seemingly very crude approximation is motivated by a  
 98 model of atoms with a uniform density inside the volume of the atoms. Our approach is correct in  
 99 two limiting cases: i) the neutral, "free" atom and ii) when the atom has no electrons.

100 In 2009, Tkatchenko *et al.* linked the dispersion coefficient for an atom in a molecule ( $C_{6,i}^{aim}$ ) to the  
 101 dispersion coefficient of the free atoms ( $C_{6,i}^{free}$ ) through the ratio displayed in eq. (4):<sup>15</sup>

102 
$$C_{6,i}^{aim} = \left( \frac{V_i^{aim}}{V_i^{free}} \right)^2 C_{6,i}^{free} \quad (6)$$

103 Directly exploiting our assumption we can define a new relation for the dispersion coefficient:

104 
$$C_{6,i}^{aim} = \left( \frac{N_i}{Z_i} \right)^2 C_{6,i}^{free} \quad (7)$$

105 We here discuss the results obtained with equation (7) with the  $C_{6,i}^{free}$  available for most of the  
 106 elements in the periodic table as provided by Grimme.<sup>17</sup>

107 We apply the same mixing rule as in our previous work<sup>25</sup> for the dispersion coefficients between  
 108 atoms  $i$  and  $j$ .

109 
$$C_{6,ij} = \frac{2C_{6,i}C_{6,j}}{C_{6,i} + C_{6,j}} \quad (8)$$

110 It is important to stress that a dispersion correction based on the Mulliken scheme is ideally suited  
 111 for SCC-DFTB. In contrast to large basis sets, small or minimal basis sets provide robust Mulliken  
 112 charges.<sup>50</sup> In this respect, DFTB Mulliken charges (using minimal basis sets) are robust and very  
 113 convenient since available at no cost in contrast to Hirshfeld charges (see details on the charges and  
 114 on the  $C_6$  coefficients in the Supplementary Materials).

115 The dDsC damping is based on the universal Tang and Toennies function<sup>51,52</sup> plus a second  
 116 damping function with both depending on the information extracted from the electron density.

117 dDMC preserves the double damping and the flexibility of dDsC but without density-dependency  
 118 and without adding extra cost to the electronic structure computation. In line with the “density  
 119 dependent” approach, the damping function uses an electronic parameter to switch the correction on  
 120 and off. The Fermi function<sup>53</sup>  $F(a, s, R_{ij})$  damps a Tang and Toennies function  $TT(b, R_{ij})$  to ensure  
 121 enough flexibility

$$f_d(R_{ij}, b) = F(a, s, R_{ij})TT(b, R_{ij}) \quad (9)$$

122 The Fermi damping function

$$F(a, s, R_{ij}) = \frac{1}{1 + \exp\left(-s\left(\frac{R_{ij}}{aR_{ij}^{vdW}} - 1\right)\right)} \quad (10)$$

125 contains an empirical parameter  $a$  that scales the van der Waals radii<sup>54,55</sup> ( $R_{ij}^{vdW} = R_i^{vdW} + R_j^{vdW}$ ) and  
 126 a steepness parameter  $s$  that minimizes the effect of the Fermi function at larger internuclear  
 127 distances. The Tang and Toennies function is

$$TT(b, R_{ij}) = 1 - e^{(-b, R_{ij})} \sum_{k=0}^6 \frac{(b, R_{ij})^k}{k!} \quad (11)$$

128 in which the TT-damping factor ( $b_{ij}$ ) regulates the medium range of the correction.  $b_{ij}$  is computed  
 129 according to the combination rule

$$b_{ij} = 2 \frac{b_i b_j}{b_i + b_j} \quad (12)$$

132 The dDsC  $b_{ii}$  factor is defined as

$$b_i = b_0 \sqrt[3]{\frac{1}{\alpha_i}} \quad (13)$$

134 where  $b_0$  is a fitted parameter and  $\alpha_i$  is the polarizability of the interacting atoms in the molecule

135 defined as  $\alpha_i = \alpha_i^{free} \frac{V_i^{aim}}{V_i^{free}}$ . Adopting the same idea to dDMC we get

$$b_{ii} = b_0 \sqrt[3]{\frac{1}{\alpha_i}} = b_0 \sqrt[3]{\frac{1}{\alpha_i^{free}}} \sqrt[3]{\frac{V_i^{free}}{V_i^{aim}}}; \quad b_0 \sqrt[3]{\frac{1}{\alpha_i^{free}}} \sqrt[3]{\frac{N_i}{Z_i}} \quad (14)$$

137 where the free atom polarizabilities ( $\alpha_i^{free}$ ) are taken from the CRC Handbook.<sup>56</sup>

138 This scheme and in particular the usage of a “double-damped” function ensures the right behaviour  
139 at both the short and medium/long range.

140 Note that akin to other dispersion corrections, -dDMC cannot solve the issues related to the poor  
141 description of H-bonded interactions by semi-empirical approaches.<sup>57,58</sup> Instead an additional  
142 empirical correction would be needed<sup>48</sup> for this purpose. However, our present objective is not to  
143 improve the description of H-bonds and increase the empiricism by adding a second correction but  
144 rather to provide a fast electronic structure scheme that accurately describes sulfur-containing  
145 compounds involved in pi-stacking interactions. Nevertheless, the proposed dDMC scheme could  
146 be further combined with a “H-bond corrections” to provide a more generally applicable scheme.

## 147 **Gradient**

148 The validation of the quality of the approximated gradients is essential in the context of both  
149 optimizations and molecular dynamic trajectories. Since the dispersion correction depends, through  
150 the Mulliken charges, on the electronic structure, the gradient has an electronic and a geometric  
151 contribution. To improve the computational efficiency, we neglect the electronic contribution, i.e.,  
152 the Mulliken charges are fixed with respect to the atom displacements. In addition, given that the  
153 dispersion correction depends only on the interatomic distances, the gradients are computed directly  
154 in function of the distance instead of the coordinate’s displacements:

$$155 \quad F_{i,x} = - \sum_{j \neq i}^N \frac{d(f_d(R_{ij})C_{6,ij}R_{ij}^{-6})}{dR_{ij}} ; - \sum_{j \neq i}^N \frac{d(f_d(R_{ij})R_{ij}^{-6})}{dR_{ij}} C_{6,ij} \cdot \frac{\partial R_{ij}}{\partial x_i} \quad (2)$$

156 where  $F_{i,x}$  is the force acting on the atom  $i$  along the direction  $x$ . The validity of this  
157 approximation is verified through a comparison between the dispersion numerical gradients  
158 (computed using a displacement = 0.001 Å) and the approximated analytical ones on all the  
159 structures available in the S66 data set.<sup>59</sup> To enforce that gradient values are different from zero, a  
160 noise corresponding to a uniformly distributed random displacement comprised between -0.2 and  
161 0.2 Å was added to the coordinates of all atoms. The Mean Average Error (MAE) computed as

162 
$$\text{MAE} = \frac{\sum_i^N |f_i - r_i|}{N} \quad (3)$$

163 using the numerical derivatives as reference values ( $r_i$ ) and the approximated analytical gradients as  
164 forecast values ( $f_i$ ) is  $6.81 \times 10^{-8}$  eV/Å on an absolute average dispersion force of  $6.44 \times 10^{-3}$  eV/Å.  
165 This implies that the error arising from the approximated analytical gradient is  $\sim 1 \times 10^{-5}$  smaller than  
166 the average dispersion force arising from a deviation of 0.2 Å from the most stable configuration.

### 167 **Adjustable Parameters and Training Set**

168 The dDMC correction depends on two adjustable parameters,  $a$  and  $b_0$ , as well as on the steepness  
169 factor,  $s$ . In line with our former work,<sup>25</sup> the steepness factor,  $s = 46$ , was chosen such as to  
170 minimize the effect of the Fermi function on the overall damping at large internuclear separations  
171 by imposing the constraint  $F(a, s, 1.1 \cdot a \cdot R_{ij}^{vdW}) \geq 0.99$ . Such a limitation turns the Fermi function off  
172 when the distance between the atoms ( $R_{ij}$ ) is larger than  $1.1 \cdot a \cdot R_{ij}^{vdW}$  so that only the  $TT(b_0, r_{ij})$   
173 damping is active in this region.

174 The two parameters,  $a$  and  $b_0$ , are trained for each electronic structure approach (the values are  
175 given in the SI) with the Nelder-Mead optimization method to reproduce a set of interaction  
176 energies. The training set includes a subset of the S66x8 data set<sup>59</sup> (namely the “dispersion” and  
177 “mixed” structures). To ensure a good performance on sulfur-containing compounds, the training  
178 set is completed with an expanded version of the Pi29n (i.e., Pi29nx8) that mimics the S66x8<sup>59</sup> data  
179 set adding 7 displaced structures for each dimer included in the original Pi29n.<sup>60</sup> For each structure,  
180 one scales the equilibrium distance between the monomers centre of mass (by 0.9, 0.95, 1.05, 1.10,  
181 1.25, 1.50, 2.00). The interaction energies for each dimer (232 structures) are estimated at the  
182 CCSD(T)/CBS level<sup>61,62</sup> and corrected for BSSE using the Counter Poise (CP) correction<sup>63</sup>  
183 following the same scheme as used in the Pi29n data set.<sup>60</sup> The inclusion of energy profiles in the  
184 training phase serves to improve the response of the damping function at small intermolecular  
185 distance. To avoid issues arising from the self-interaction error, the charge transfer TTF-TCNQ



186 complex originally presents in the Pi29n data set<sup>60</sup> was not considered herein. In addition, since the  
187 emphasis is placed on improving the description of  $\pi$ -stacked motifs that are prevalent in organic  
188 electronic materials, we did not consider the hydrogen-bond complexes of the S66x8 data set. Note  
189 finally that the PM6 Hamiltonian<sup>64</sup> that was used for comparisons required the training of one  
190 additional scaling factor for the total dispersion energy.

## 191 **Computational Details**

192 All the computations are performed using a modified version of the ASE package<sup>65</sup> that applies the  
193 dDMC and D3 dispersion corrections to PM6 and DFTB3. The DFTB3 computations are performed  
194 using the DFTB+ 1.2.1 software<sup>66</sup> with the 3OB Slater-Koster files.<sup>41,42</sup> The Hubbard parameters,  
195 their derivatives as well as the  $\gamma$  factor are chosen according to Qui *et al.*<sup>41,42</sup>. PM6 computations  
196 were performed in MOPAC2012<sup>67</sup> with the default options. The D3 parameters are taken from  
197 Rezac *et al.*<sup>48</sup> and used with the software available from Grimme's website. The dDMC correction  
198 terms were computed using a standalone code that is distributed freely.

199 The ASE package delivers the QuasiNewton method to perform the geometry optimizations using  
200 homemade interfaces with the previous cited software. Optimizations are considered as converged  
201 if the forces on all individual atoms are below  $5 \times 10^{-3}$  eV/Å.

202 The Born-Oppenheimer molecular dynamics uses the implementation of the velocity Verlet  
203 algorithm in ASE to integrate the DFTB3/3OB and the corrected trajectories. The PBE-dDsC  
204 simulations are performed with a modified version of the QCHEM4.0 software package.<sup>68</sup>

205 The DFTB3/3OB simulations involving the dithiacyclophane molecule are performed in the NVE  
206 ensemble using a 0.5 fs time step in line with the PBE and PBE-dDsC trajectories obtained  
207 previously by Corminboeuf and co-workers.<sup>69</sup> The starting structures were the same as in Ref. <sup>69</sup>.  
208 The initial temperature was set to 300 K. With this condition no energy drift has been observed. To  
209 ensure that the approximate gradients do not introduce instability in the MD trajectory, we used  
210 dithiacyclophane to verify what is the maximum time step that does not introduce an energy drift.

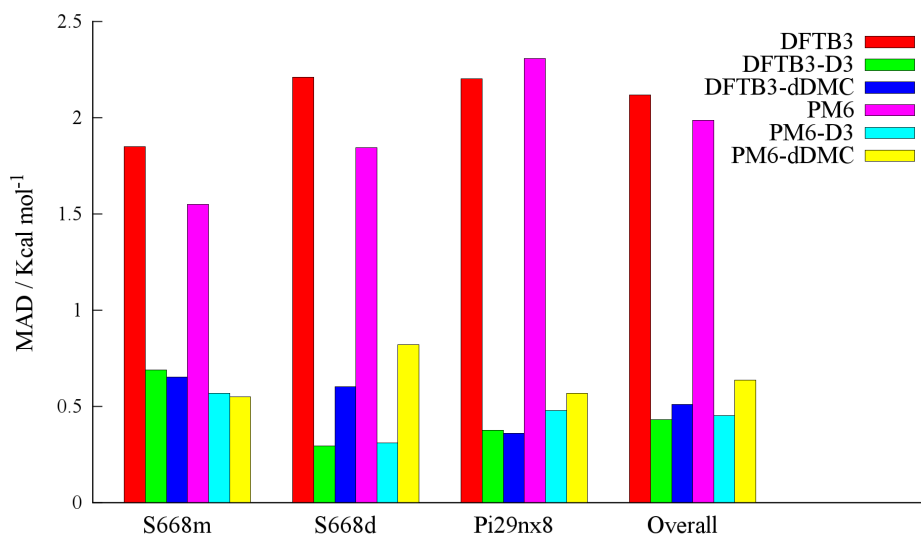
211 The drift was defined as the angular coefficient of the trendline that best fits the energy profile as a  
212 function of time. We computed the drift on the NVE simulations starting at 300 K from the same  
213 initial structure. Each simulation lasts 10 ps for both DFTB3/3OB and DFTB3/3OB-dDMC. The  
214 result shows that the approximate gradient does not influence the energy conservation since both  
215 methods show a critical drift for a time step of 2.4 fs.

216 The simulations on the caffeine-receptor complex<sup>70-72</sup> are conducted in the NVT ensemble using the  
217 Langevin thermostat with a  $2 \text{ ps}^{-1}$  friction. We found that a 1 fs time step is small enough to avoid  
218 energy drifts. The so-called “reference structure”, optimized at the PBE0-dDsC/def2-SVP<sup>70</sup> level  
219 was taken as the initial structure for the DFTB3/3OB simulations. A snapshot of the DFTB3/3OB  
220 trajectory after 5 ps was taken as the starting structure for the PBE-dDsC simulation. All trajectories  
221 were thermalized for 5 ps at their respective level.

## 222 **Results**

223 The following illustrates the performance of dDMC not only on interaction energies of static dimer  
224 structures but especially on practical examples featuring geometry optimizations and molecular  
225 dynamics simulations.

226 Figure 1 displays the mean absolute error of the dDMC correction applied to DFTB3/3OB and PM6  
227 compared to the uncorrected variants and the -D3 corrected energies.



228

229 **Figure 1:** Mean absolute error of DFTB3/3OB<sup>42</sup> and PM6 and the D3<sup>48,73</sup> and dDMC dispersion  
 230 corrected variants. The “overall” data set displays the MAD of all the dataset together.

231 Overall, the mean absolute errors for the DFTB3 corrected energies are below 0.7 kcal mol<sup>-1</sup>.

232 Despite its simplicity, the performance of -dDMC is very similar to -D3 except for the S66\*8d

233 subset that is notably better described at the DFTB3-D3 level (MAD = 0.29 kcal mol<sup>-1</sup>). The poorer

234 performance of dDMC for this specific subset essentially arises from an overbinding of the

235 hydrogen-rich dimers such as those made of aliphatic chains (e.g., neopentane, pentane as seen in

236 the SI). The present focus is essentially placed on  $\pi$ - $\pi$  stacking but the combination of -dDMC with

237 a H4<sup>48</sup> type correction that contains a specific repulsive term to correct the interaction between

238 hydrogen atoms would surely improve the results for these complexes. The superior performance of

239 DFTB3-dDMC as compared to PM6-dDMC is rooted in the less reliable Mulliken charges

240 associated with the PM6 Hamiltonian.<sup>64</sup> Besides the reasonable performance, a clear benefit of

241 using the dDMC scheme is certainly the gain in computational speed, which is especially visible

242 when computing the gradients on large-scale systems (80% faster than the DFTD3 program version

243 2.1 rev 3 in calculating gradients on around 1000 atoms, see Supplementary Materials).

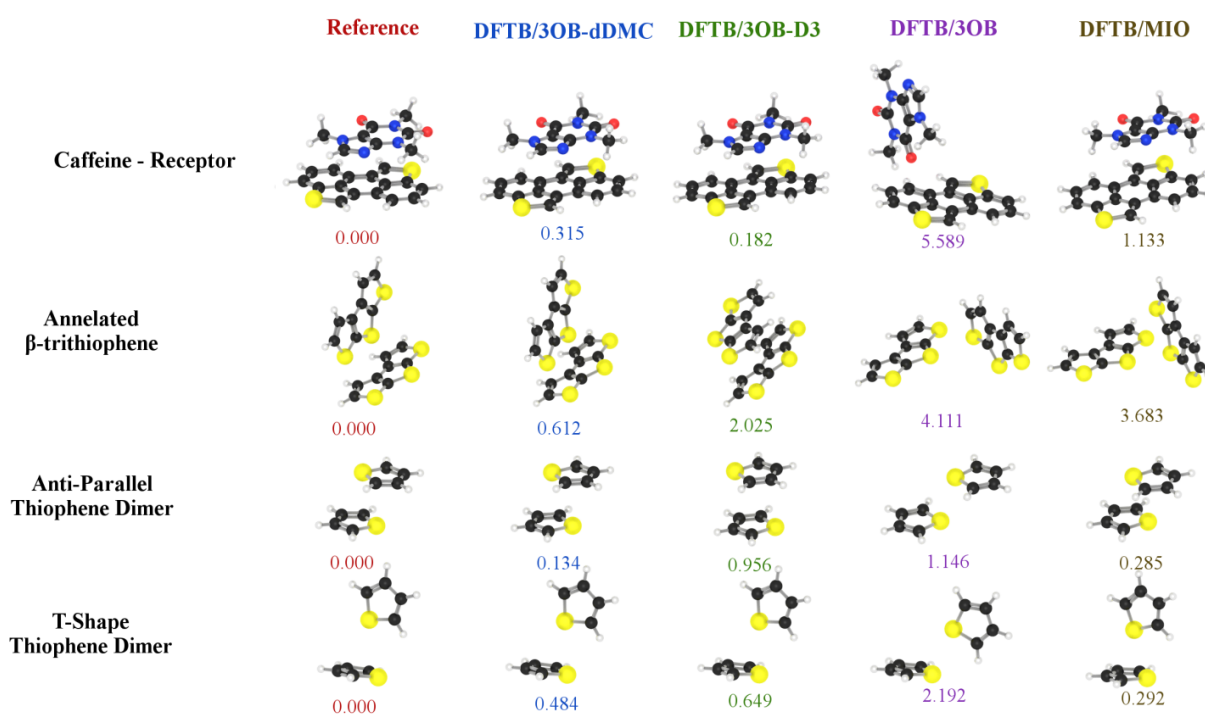
244 The authors recently demonstrated that the original DFTB3/MIO11<sup>44</sup> parameters lead to spurious

245 energies and geometries for any systems that features a sulfur atom involved in a non-covalent

246 interaction.<sup>45</sup> As illustrated by the examples provided in Figure 2, the latest 3OB parameterization

247 by Cui *et al.*<sup>41,42</sup> offers a dramatic improvement over MIO11 for the dispersion corrected gas phase  
 248 geometries. DFTB/3OB-dDMC leads to four geometries that are in close agreement with the  
 249 reference PBE0-dDsC/def2-SVP<sup>71</sup> or RI-MP2/TZ<sup>74-76</sup> data. In particular, the T-shape thiophene  
 250 dimer and the illustrative caffeine-receptor comple<sup>70,71</sup> are well reproduced with both DFTB/3OB-  
 251 dDMC and DFTB/3OB-D3 (RMSD < 0.4 Å). The -D3 description of the annulated β-trithiophene  
 252 dimer<sup>45,77</sup> converges toward another minima (RMSD= 2.025 Å), whereas dDMC remains in  
 253 agreement with the reference data (RMSD = 0.6Å). Similar discrepancies are observed for the anti-  
 254 parallel thiophene dimer. Note that our training set placing more emphasis on improving the  
 255 treatment of sulfur interactions could be at the origin of this difference. While the spurious  
 256 overbinding characteristic of the non-dispersion corrected DFTB/MIO11 geometries is recurrent  
 257 and relatively large in magnitude (sometime even larger than the reference interaction energy), the  
 258 3OB parameters offer a significant improvement: the DFTB3/3OB optimized geometries still bind  
 259 but the interaction energies involved are nevertheless much smaller than the reference and  
 260 dispersion corrected values.

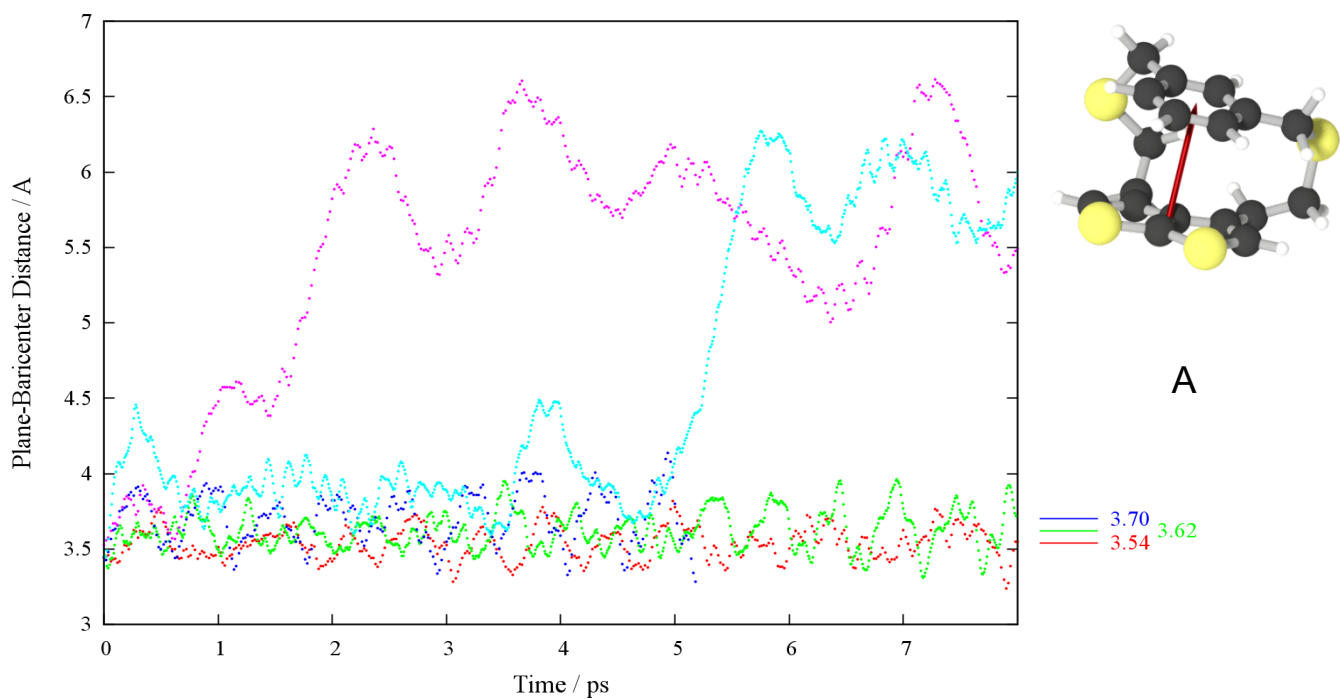
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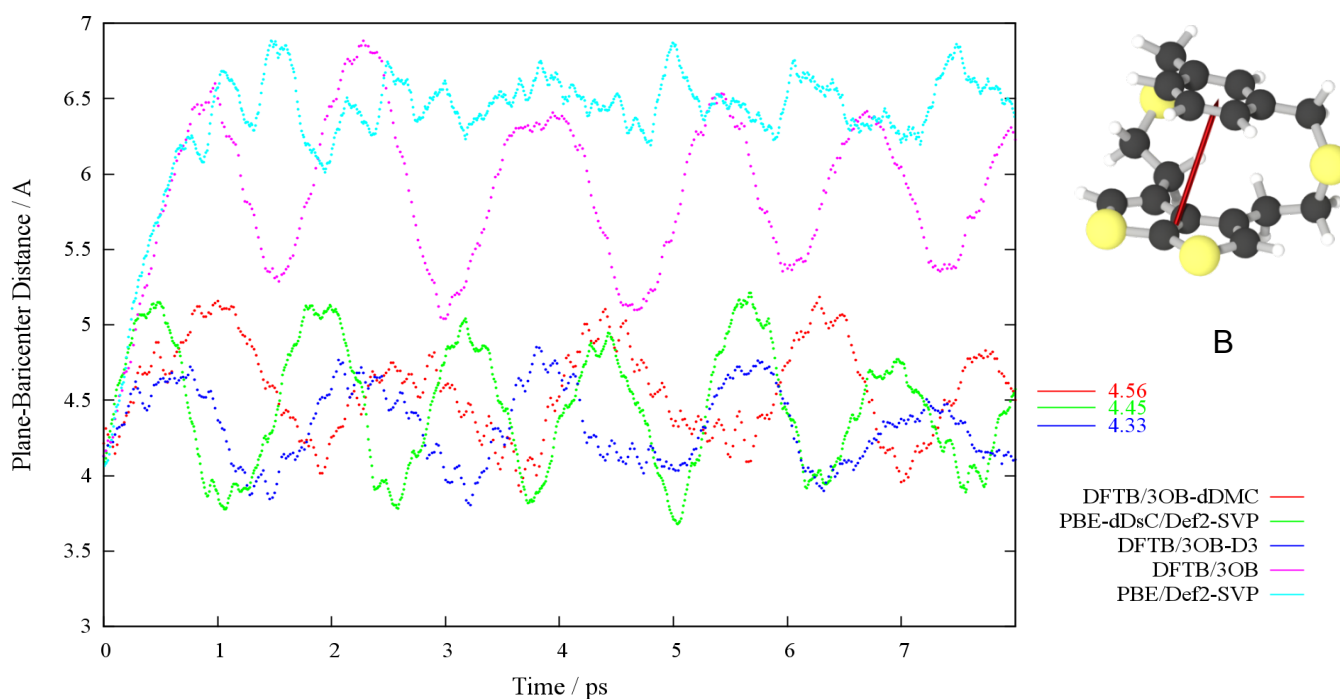
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263 **Figure 2:** Comparisons of the DFTB geometry of sulfur-containing compounds geometries at both  
264 dispersion- (dDMC and D3) and non- corrected levels (with the 3OB and MIO parameters). RMSD  
265 (Å) with respect to the reference geometry are reported for each complex. The reference geometries  
266 of the caffeine-receptor and the annelated  $\beta$ -trithiopene<sup>45,77</sup> complexes are computed at the PBE0-  
267 dDsC/def2-SVP level.<sup>71</sup> The anti-parallel thiophene dimer system is optimized at the RI-MP2/TZ<sup>74-</sup>  
268 <sup>76</sup> level with Counter Poise correction to avoid BSSE<sup>63</sup> in Turbomole5.1.<sup>78</sup>

269 The performance of DFTB3/3OB-dDMC is further validated on the Born-Oppenheimer molecular  
270 dynamic simulations of two examples dominated by intra- and intermolecular interactions  
271 respectively. Figure 3 shows MD trajectories of an illustrative dithiacyclophane incorporating a  
272 thieno-[2,3-b]-thiophen that was originally chosen to evaluate the importance of self-consistency in  
273 dDsC.<sup>69</sup> This compound is rather challenging due to its large flexibility inherent to the existence of  
274 several low energy conformers featuring both  $\pi$ - $\pi$  stacked and open conformations. We here present  
275 the molecular dynamic trajectories starting from two closed (i.e.,  $\pi$ - $\pi$  stacked) conformers (**A** and  
276 **B**) and monitor the distance between the barycentre of the benzene ring and the middle of the C-C  
277 bond of the thienothiophene ring over 8 ps trajectories that are directly compared to our former  
278 PBE-dDsC simulations taken from Corminboeuf and co-workers<sup>69</sup>. On average, the three tested  
279 dispersion-corrected schemes lead to very similar distances with no systematic  
280 over/underestimation: The average distance at the DFTB/3OB-dDMC level is the longest for the  
281 first trajectory (i.e., starting from conformer **A**) but the shortest for the second trajectory. The  
282 simulation of the closed conformer **B** performed at both non-corrected levels, PBE and DFTB/3OB,  
283 readily open. In contrast, the opening of conformer **A** differs significantly between PBE and  
284 DFTB/3OB. The PBE opening process is relatively sudden, whereas the DFTB/3OB structure opens  
285 more gradually. Other deviations observed between the non-corrected approaches include the larger  
286 flexibility of DFTB as compared to PBE.



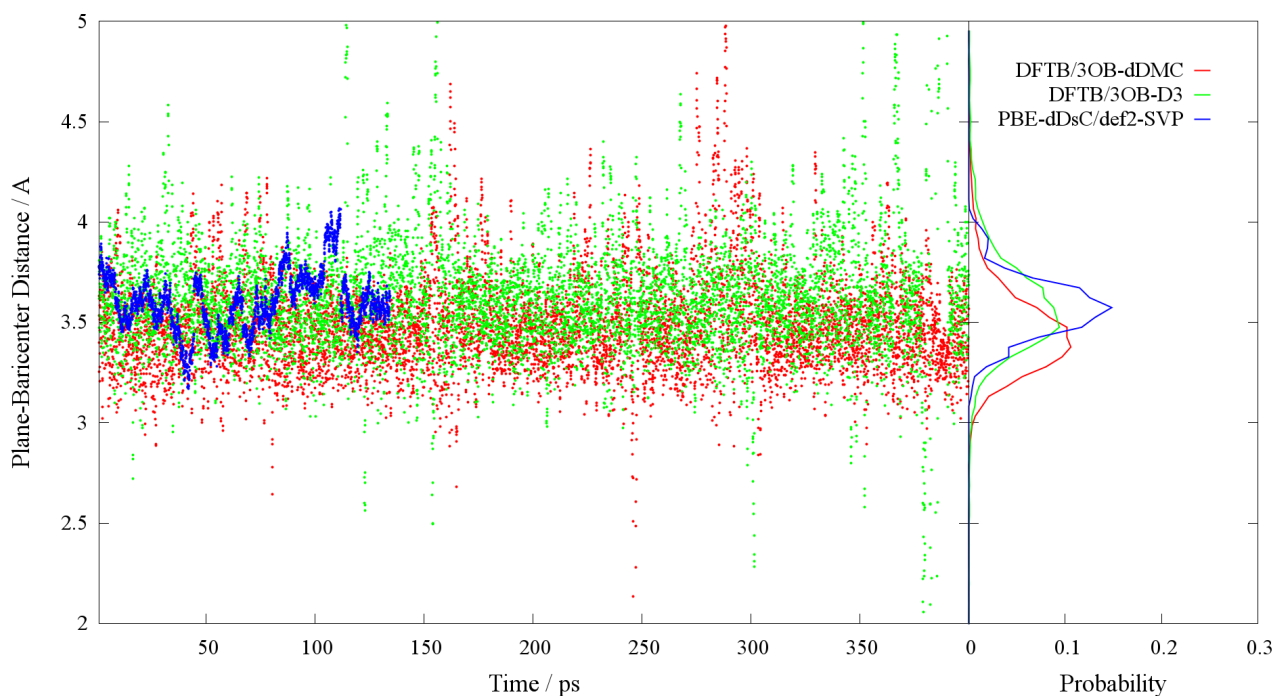
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288

289 **Figure 3:** Profiles of the distance between the barycenter of the benzene ring and the middle of the  
 290 C-C bond of the thienothiophene ring over a NVE Born-Oppenheimer molecular dynamic trajectory  
 291 (see red arrow). Computations are performed at room temperature with non-corrected and  
 292 dispersion corrected PBE and DFTB/3OB. The starting structure are optimized at  $\omega$ B97X-D/6-  
 293 31G\* level<sup>79-81</sup>. The methods used to perform the simulations are distinguished by color.

294 The last molecular dynamics example (Figure 4) inspects the longer range intermolecular  
295 interaction of an illustrative caffeine-receptor dimer already studied in Refs<sup>[71,72]</sup> that is also  
296 included in our comparisons of geometry optimizations (Figure 2). The distance monitored is taken  
297 between a plane that best incorporates the atoms of the receptor and the barycenter of the caffeine  
298 molecule. The 350 ps DFTB3/3OB-dDMC trajectory is compared to that of DFTB3/3OB-dDMC  
299 and to a shorter 134.5 ps PBE-dDsC/def2-SVP trajectory. The histogram and overall trajectories  
300 show a nice correlation between the two DFTB/3OB corrected approaches correlate. The overall  
301 observation is that accounting for dispersion is ubiquitous when performing *ab initio* molecular  
302 dynamic trajectories. Our simulations also demonstrate that the residual error related to the 3OB  
303 sulfur parameters is counterbalance if combining DFTB/3OB with a dispersion correction. Within  
304 the framework of DFTB, dDMC represents a very simple and efficient alternative to the existing  
305 schemes specifically adapted to biomolecules for addressing problematic relevant to the field of  
306 organic electronic.



307

308 **Figure 4:** Profiles and histograms of the distance between the average plane of the receptor and the  
309 the barycentre of the caffeine over a NVT Born-Oppenheimer molecular dynamic trajectory.

310 Computations are performed at room temperature with DFTB3/3OB-dDMC, DFTB3/3OB-D3 and  
311 PBE-dDsC/def2-SVP.

## 312 **Conclusions**

313 This work introduces a fast atom-pairwise dispersion correction based on Mulliken charges that is  
314 specifically tailored for DFTB3. Unlike previous dispersion corrected DFTB computations focusing  
315 on biological systems, we here place a special emphasis on improving the description of compounds  
316 prevalent in the field of organic electronics. In this respect, the dDMC scheme performs especially  
317 well for the energies (MAD=0.7 kcal mol<sup>-1</sup> for the test set of 94 compounds for a total of 752  
318 different systems), geometries and molecular dynamics of sulfur-containing moieties involved in  $\pi$ -  
319  $\pi$  stacking interactions that are known to be especially challenging for DFTB. We have thus  
320 provided both, a valuable extension to DFTB3 by providing a charge-dependent dispersion  
321 correction and a careful validation of the provided scheme on test sets for typical weak interactions  
322 (S66) and motives typical for organic electronics (Pi29).

323 The rising interest in organic electronic materials along with the simplicity of the proposed  
324 correction suggests that this approach has great potential. Future developments should enable the  
325 treatment of explicit solvent and the consideration of many body contributions that potentially play  
326 a role in determining the geometries and thermodynamics of nanoscale assemblies of organic  
327 molecules.

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330 **SUPPORTING INFORMATION** Additional Supporting Information may be found in the online  
331 version of this article.

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